

Sinter Crystallization of Soda Lime Silicate Waste Glass Modified With Eggshell-Derived CaO for Glass Ceramic Tile Applications

Abstract

4 *The utilization of post-consumer and industrial wastes for the production of glass ceramics*
5 *provides a sustainable pathway for developing high performance construction materials. In this*
6 *study, glass-ceramic tiles were synthesized via the sinter crystallization technique using waste*
7 *soda lime silicate (TV panel) glass and eggshell-derived calcium oxide (CaO) as alternative raw*
8 *materials. The waste glass served as a silica rich matrix, while eggshell waste acted as a low*
9 *cost and sustainable CaO source. Raw materials and sintered products were characterized using*
10 *X-ray fluorescence (XRF), X-ray diffraction (XRD), Fourier transform infrared spectroscopy*
11 *(FTIR), and scanning electron microscopy coupled with energy dispersive spectroscopy*
12 *(SEM/EDS). Compacted samples were sintered at 650, 700, and 750 °C, and their phase*
13 *composition, microstructure, density, mechanical properties, abrasion resistance, and chemical*
14 *durability were evaluated. XRD analysis revealed partial crystallization within a residual glassy*
15 *matrix, with quartz and calcite identified as the dominant crystalline phases. Increasing the*
16 *sintering temperature promoted crystallization and densification up to 700 °C, as confirmed by*
17 *SEM observations. The sample sintered at 700 °C exhibited optimal performance, achieving the*
18 *highest compressive strength (28.54 MPa), enhanced density, improved hardness, superior*
19 *abrasion resistance, and excellent chemical durability in acidic and alkaline environments.*
20 *These results demonstrate that waste glass and eggshell derived CaO can be effectively utilized*
21 *to produce durable glassceramic tiles suitable for sustainable building applications.*

22 **Keywords:** Characterization, Eggshellderived CaO, Glass ceramics, Mechanical properties, Sinter
23 crystallization, Waste glass.

24 1. Introduction

25 The growing demand for sustainable construction materials has intensified research into the
26 valorization of industrial and post-consumer wastes as alternative raw materials for advanced
27 ceramics. Although nature has provided essential components for the manufacture of glasses with
28 abundant and easily removable components for centuries, synthetic chemical products and a
29 wide range of solid waste are also used today (Rawlings *et al.*, 2006). Waste recycling has
30 become an unavoidable requirement to optimize the use of natural resources (Rincon, 2016).
31 Furthermore, by using wastes as alternative raw materials, it is possible to obtain new materials
32 with added value and mitigate the environmental risk (Rincon, 2016; Barbieri *et al.*, 2000;
33 Saparuddin *et al.*, 2020). However, the use of waste as raw material requires strict control over
34 the chemical composition, since the mixtures must contain the typical components of stable

glasses, in particular Si and Al oxides and nucleating agents that favor crystallization. In this way, each residue must contribute an appropriate amount of vitrifying agents such as SiO_2 and Al_2O_3 , as well as elements that modify and promote fusion ($\text{Na}_2\text{O}_3\text{K}_2\text{O}$), in addition to stabilizing agents such as CaO and MgO , which are the components that give rise to glasses with adequate or special characteristics (Barbieri *et al.*, 2000; Karamanov *et al.*, 2003). Among these wastes, soda lime silicate glass constitutes a major fraction of municipal solid waste streams due to its extensive use in packaging, glazing, and electronic applications. Although glass is theoretically recyclable, large quantities of waste glass remain underutilized because of contamination, color mixing, and economic constraints, leading to significant environmental burdens (Yang *et al.*, 2014). Converting waste glass into value added glass ceramic products represents an effective strategy to mitigate disposal challenges while producing materials with enhanced functional properties.

Glass ceramics are polycrystalline materials produced through the controlled crystallization of parent glasses and are characterized by a combination of crystalline phases embedded within a residual glassy matrix (Deubener *et al.*, 2018). This unique microstructure enables glass ceramics to exhibit superior mechanical strength, chemical durability, abrasion resistance, and thermal stability compared with conventional glasses (Holand & Beall, 2019). Consequently, glass ceramics have found widespread applications due to their characteristics, such as chemical and mechanical durability, surface hardness, low thermal expansion coefficient, which are easily adapted to practical uses in architectural tiles, flooring materials, countertops, and other load bearing construction components (Mauro & Zanotto, 2014). Among the various processing routes, the sinter crystallization technique has attracted considerable attention for waste derived glassceramics, as it allows densification and crystallization to occur simultaneously from compacted glass powders at relatively lower temperatures, reducing energy consumption and processing costs (Karamanov & Pelino, 2006; Bernardo & Colombo, 2006).

The incorporation of calcium rich additives plays a critical role in tailoring the crystallization behavior and properties of soda lime silicate glass systems. Calcium oxide (CaO) acts as a network modifier, influencing viscosity, nucleation kinetics, and phase development during heat treatment (Shelby, 2005; Zanotto & Mauro, 2017). Conventionally, CaO is introduced using high-purity chemical reagents or natural limestone; however, these sources are associated with

65 additional cost and environmental impacts. In this context, waste eggshells have emerged as an
66 attractive, low-cost, and sustainable alternative CaO source. Eggshells are primarily composed of
67 calcium carbonate (CaCO_3), which decomposes to CaO upon heating, and are generated in large
68 quantities by households, restaurants, and food-processing industries. Their reuse not only
69 diverts biodegradable waste from landfills but also aligns with circular economy
70 principles(Onwubu *et al.*, 2019; Arshad *et al.*, 2021).

71 Recent studies have demonstrated the feasibility of utilizing eggshell derived CaO in
72 cementitious materials, bio ceramics, and glass formulations, where it contributes to improved
73 densification, phase formation, and mechanical performance(Onwubu *et al.*, 2019; Arshad *et al.*,
74 2021). However, investigations focusing on the modification of waste soda lime silicate glass
75 with eggshell derived CaO for glass ceramic tile applications remain limited. In particular, there
76 is a need to better understand how CaO addition influences sinter crystallization behavior, phase
77 evolution, microstructure, and key performance indicators such as mechanical strength and
78 chemical resistance at relatively low sintering temperatures.

79 Therefore, the present study investigates the sinter crystallization of soda lime silicate waste
80 glass modified with eggshell derived CaO for the development of glass ceramic tiles. Emphasis
81 is placed on the role of eggshell derived CaO in promoting crystallization, microstructural
82 refinement, and durability of the resulting glass ceramics. By combining two abundant waste
83 streams, waste glass and eggshells this work aims to demonstrate a sustainable and cost effective
84 route for producing high performance glass ceramic materials suitable for tiling and other
85 building applications

86

87 **2.0 Materials and Methods**

88 **2.1 Raw Materials**

89 Waste eggshells and soda lime silicate television panel glass were employed as the primary raw
90 materials. The waste TV panel glass was sourced from discarded and damaged television screens
91 collected from local television repair workshops within Funtua Local Government Area, Katsina

92 State, Nigeria. This type of glass represents a significant component of electronic waste (e-
93 waste) streams, as it is commonly generated during screen replacement and repair activities and
94 is often disposed of without formal recycling pathways. The eggshells were obtained from
95 chicken restaurants in Funtua, where large quantities are generated daily as food processing
96 waste. Polyvinyl acetate (PVAc) was employed as a temporary binder. Prior to processing, all
97 raw materials were thoroughly washed with distilled water to remove surface contaminants.

98 **2.2 Sample Preparation**

99 The eggshells were cleaned, air dried for 24 h, and oven dried at 110 °C for 3 h to eliminate
100 residual moisture. The dried eggshells were crushed and ground into powder. Waste soda lime
101 silicate glass were also crushed using a jaw crusher and subsequently milled in a planetary ball
102 mill for 24 h at 50 rpm to obtain a homogeneous fine powder. Both powders were sieved to a
103 particle size of 75 μm , consistent with prior studies indicating enhanced sintering and
104 densification behavior at reduced particle sizes.

105 **2.3 Characterization of Raw Materials**

106 Chemical composition of the eggshell and soda lime glass powders were determined using
107 Energy Dispersive X-ray Fluorescence (EDXRF) (Thermo Scientific QUANT'X 9952120),
108 following ASTM D8064-16. Phase identification and structural analysis were carried out by X-
109 ray Diffraction (XRD) using a Rigaku Miniflex diffractometer with Cu K α radiation ($\lambda = 1.5418$
110 \AA), operated at 30 kV and 15 mA, over a 2 θ range of 5-70°, in accordance with ASTM E3294-
111 22. Microstructural features and elemental distribution were examined using Scanning Electron
112 Microscopy coupled with Energy Dispersive Spectroscopy (SEM/EDS) (Phenom ProX),
113 following the ASTM E1508-12a. Functional groups and bonding characteristics were
114 investigated using Fourier Transform Infrared Spectroscopy (FTIR), in accordance with ASTM
115 E1252-98.

116 **3.0. Results and Discussion**

117 **3.1 Results**

118 Table 1 present the XRF results of the eggshell powder (ES) and waste glass (WG). The eggshell
119 powder is dominated by CaO (85.84 wt. %), inorganic impurities and traces were also observed
120 in the Eggshell powder. This high lime CaO content agrees with previous studies reporting
121 eggshells as calcium rich materials suitable for ceramic applications (Arshad *et al.*, 2021;
122 Shiferaw *et al.*, 2019). The waste soda lime silicate glass is characterized by a high SiO₂ content
123 (61.04 wt. %), together with modifiers and intermediate oxides. These results are consistent with
124 earlier reports on waste container and TV screen glasses (Diaz *et al.*, 2015; Kazmi *et al.*, 2021).
125 The combined CaO rich eggshell and silica rich waste glass compositions are favorable for glass
126 ceramic formation via sinter crystallization.

127 **Table1. Shows the Results of waste Soda Lime Silica Glass Using of X-ray Fluorescence**

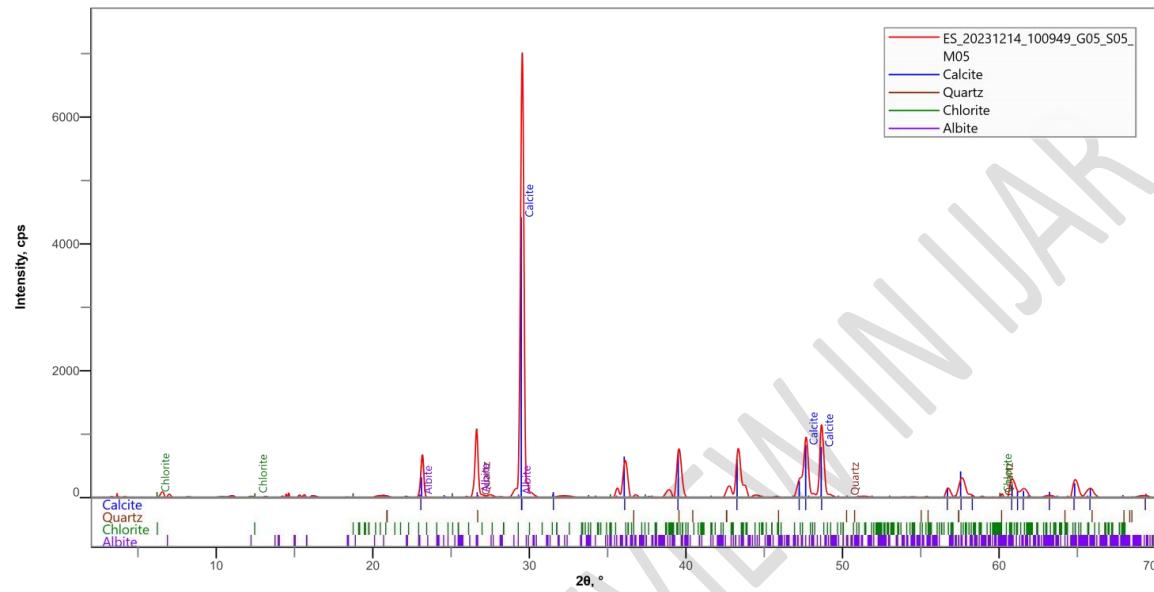
Elements (Wt. %)	SiO ₂	Na ₂ O	CaO	K ₂ O	Al ₂ O ₃	MgO	Fe ₂ O ₃	PbO	BaO	SrO	Trace
Waste glass	61.04	13.25	2.14	2.81	2.53	2.96	0.17	2.50	1.90	ND	9.17
Eggshell	1.24	3.16	85.84	0.26	2.39	2.58	0.17	ND	0.46	3.46	0.43

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129 **3.2. Microstructural analysis**

130 XRD analysis of the eggshell powder revealed calcite (CaCO₃) as the dominant crystalline phase,
131 with characteristic peaks at 20, 29°, 47°, and 48°, alongside minor phases of chlorite, albite, and
132 quartz as presented in figure 1. Quantitative phase estimation indicated approximately 80%
133 calcite, confirming eggshell as a calcium rich precursor, consistent with literature reports. In
134 contrast, the waste soda lime glass exhibited a broad diffuse halo without sharp diffraction peaks,
135 confirming its amorphous nature, which is typical of glassy materials as depicted in figure
136 2. FTIR spectra of the eggshell powder presented in figure 3. Showed prominent carbonate-
137 related bands at 1401, 715, and 872 cm⁻¹, corresponding to CaCO₃, as well as O-H vibrations
138 associated with adsorbed moisture. The waste glass FTIR spectrum was dominated by silicate
139 network vibrations, with bands at 760 cm⁻¹ (Si-O) and 969 cm⁻¹ (Si-O-Si asymmetric stretching),
140 confirming its silicate based composition. These observations agree well with previous studies by
141 Onwubu *et al.*, (2019) who conducted their study on In vitro evaluation of nanohydroxyapatite
142 synthesized from eggshell waste in occluding dentin tubules. SEM micrographs of the eggshell
143 powder revealed irregular, porous particles with non-uniform size distribution, while EDS
144 confirmed calcium as the major element, with minor Na, Al, Mg, and trace elements. The waste

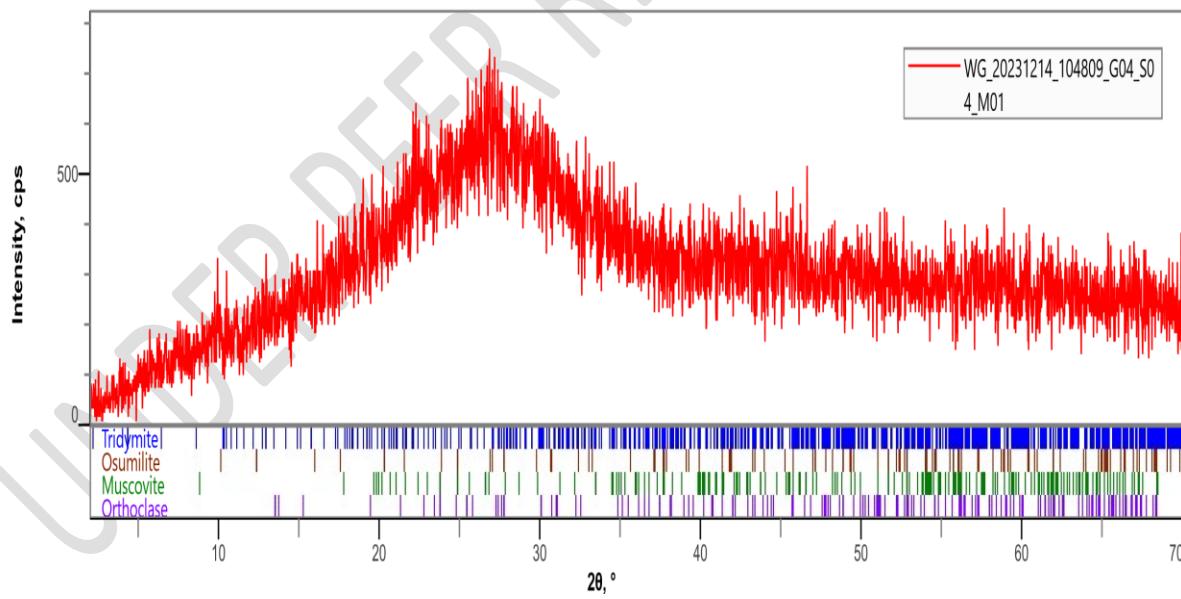
145 glass powder displayed angular and fragmented particles typical of mechanically crushed
146 amorphous glass; EDS analysis showed Si as the dominant element, with Na, Ca, Al, Mg, and K
147 as secondary constituents.



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Figure 1: X-ray Diffraction of Eggshell Powder



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Figure 2: X-ray Diffraction of Glass Powder

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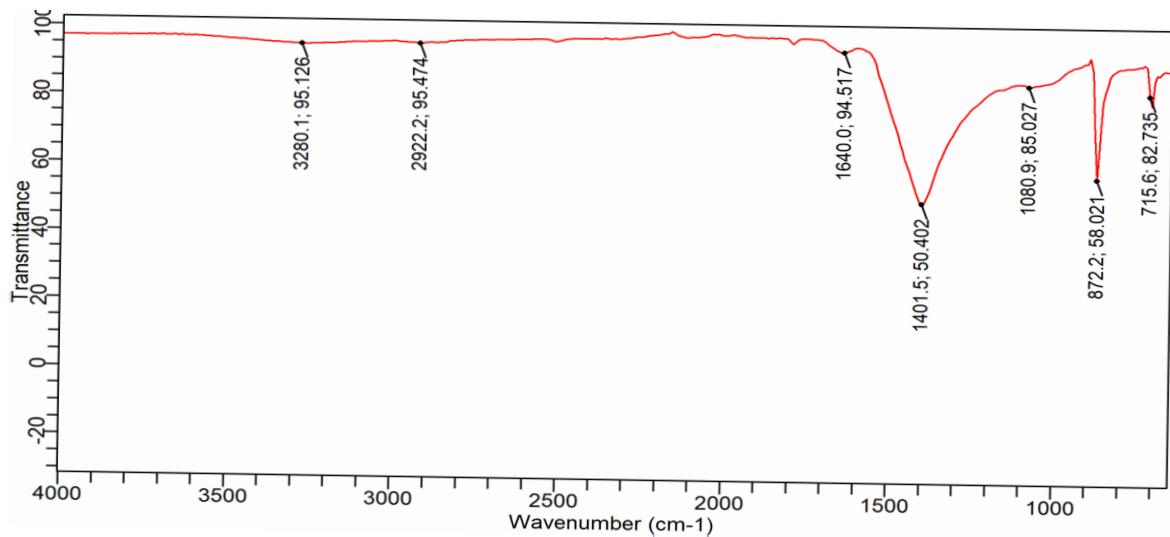


Figure 3: FT-IR of Eggshell Powder

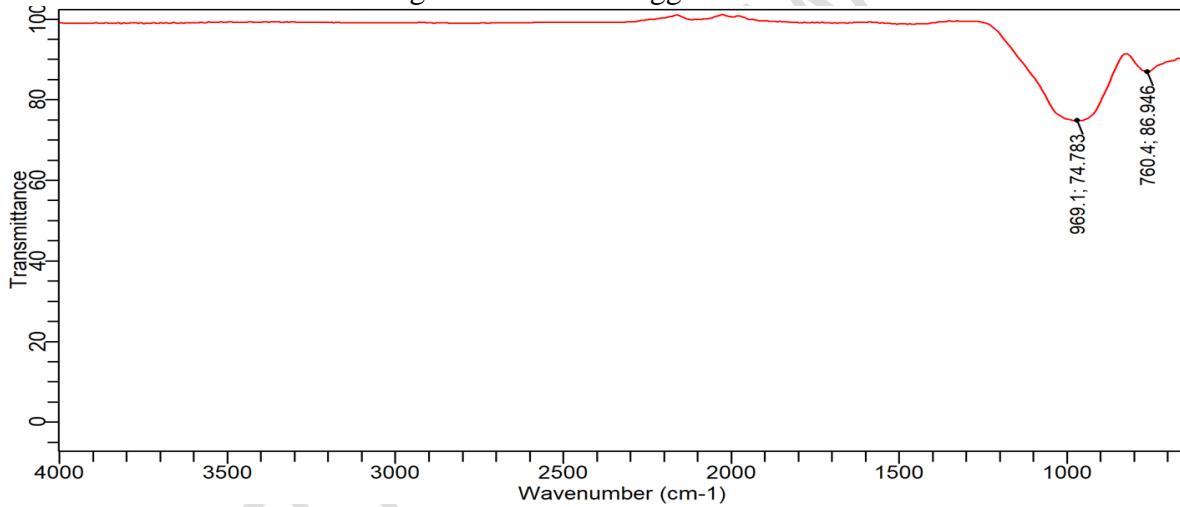
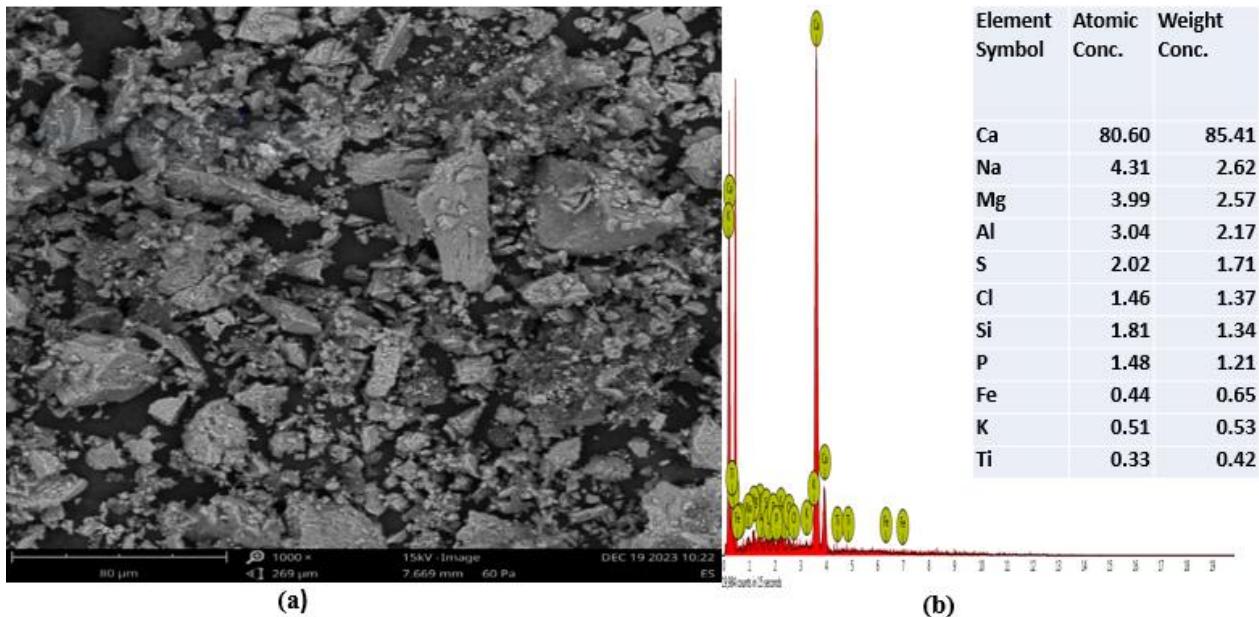


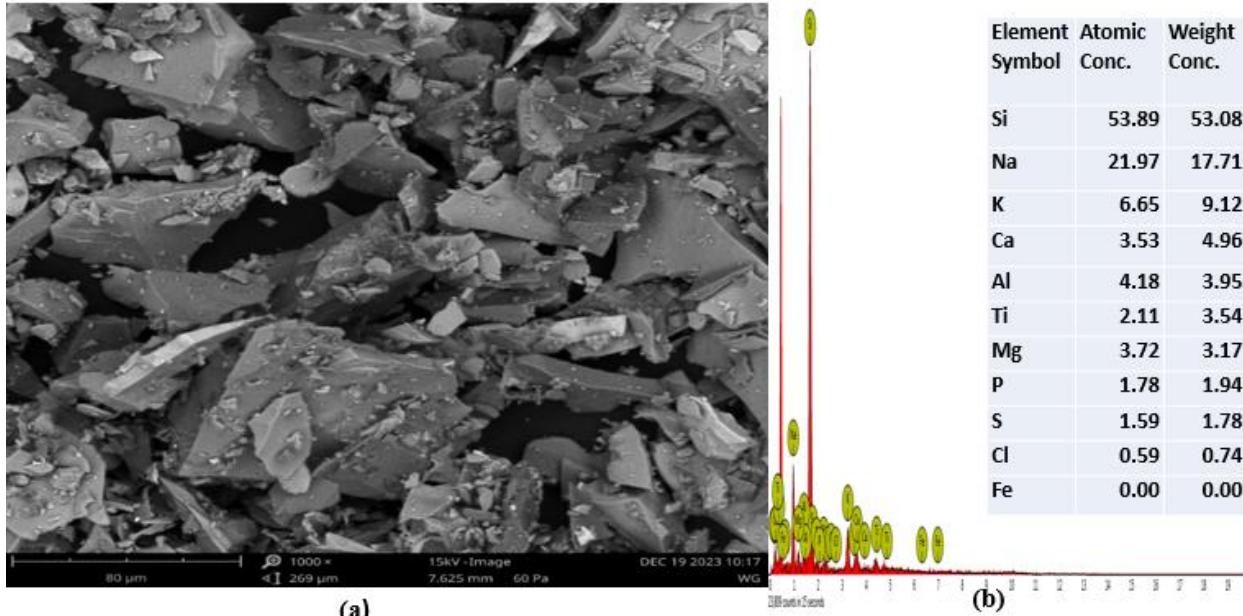
Figure 4: FT-IR of Waste Glass Powder



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Figure 5: (a) SEM (b) EDS micrographs of Eggshell powder



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Figure 6: (a) SEM (b) EDS micrographs of Waste Glass

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165 **3.3. Characterization of Glass Ceramics Samples**

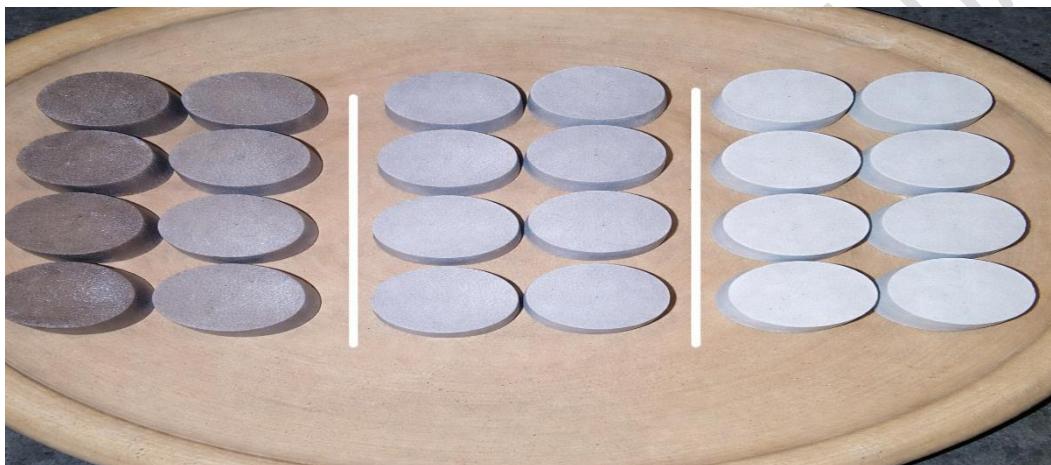
166 **3.3.1 X-ray Diffraction (XRD) Analysis**

167 The crystalline phases developed in the glass ceramic samples during controlled sinter
168 crystallization were identified by X-ray diffraction (XRD) using a Rigaku Miniflex
169 diffractometer with Cu K α radiation over a 2 θ range of 5 -70°. Phase identification was
170 performed using ICDD PDF-4 database files. The XRD patterns revealed that the sintered glass
171 ceramics consisted of a predominant amorphous matrix with superimposed crystalline peaks,
172 indicating partial crystallization. The diffraction patterns showed the presence of quartz (SiO₂) as
173 the main crystalline phase, while wollastonite (Ca₃Si₃O₉), which is commonly reported in CaO
174 SiO₂ systems, was not detected. The absence of wollastonite is attributed to the high SiO₂
175 content of the waste glass, the relatively low CaO contribution from eggshell derived additives,
176 and the comparatively low sintering temperatures employed. Similar observations have been
177 reported in previous studies, where wollastonite formation required higher CaO contents and
178 sintering temperatures close to 900 °C (Soares *et al.*, 2018). At 650 °C, only weak diffraction
179 peaks corresponding to CaCO₃ were observed, indicating limited crystallization within an
180 amorphous glassy phase. Increasing the sintering temperature to 700 °C resulted in the
181 appearance and growth of calcite (CaCO₃) as the dominant crystalline phase. Further heating to
182 750 °C led to enhanced crystallization, characterized by increased intensities of calcite and
183 quartz peaks and a reduction in the amorphous background. These temperature-dependent
184 crystallization trends are consistent with reports by (Salman *et al.*, 2017; Muganiet *et al.*, 2015) for
185 similar glass-based systems.

186 Semi quantitative phase analysis revealed that the relative contents of quartz, calcite, muscovite,
187 and orthoclase varied systematically with sintering temperature, with quartz content increasing at
188 higher temperatures. The overall degree of crystallinity was evaluated using peak area analysis
189 based on the ratio of crystalline to total (crystalline + amorphous) contributions, confirming that
190 increasing sintering temperature promotes crystallization while retaining a residual glassy phase.
191 This controlled phase evolution is essential for tailoring the microstructure and performance of
192 glass ceramics intended for tile applications.

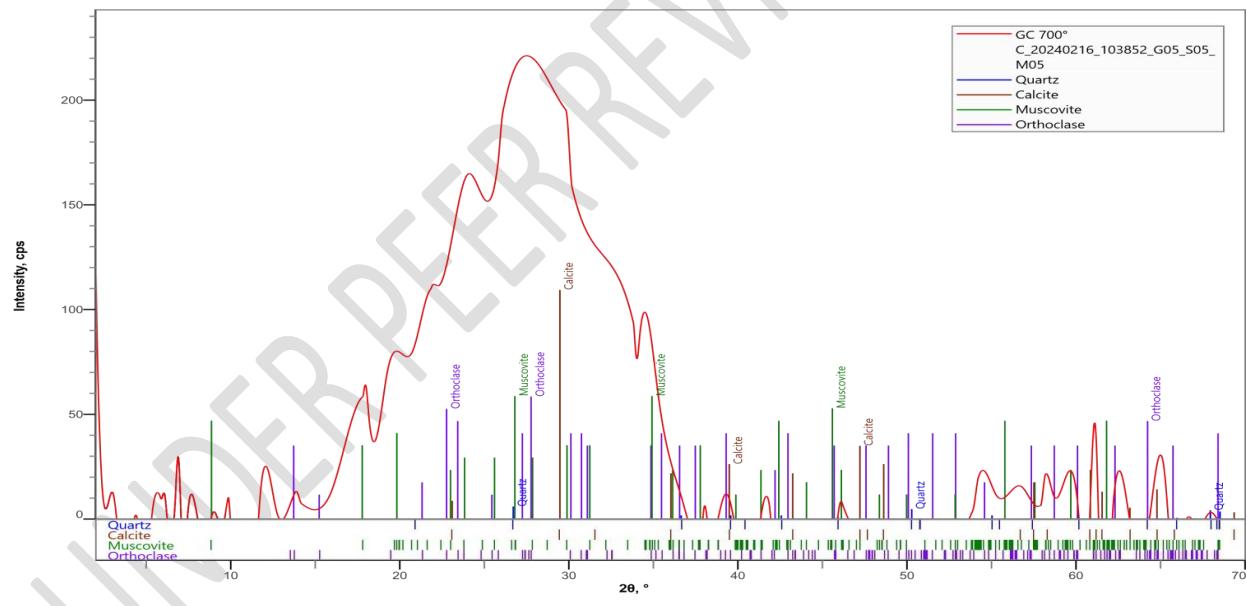
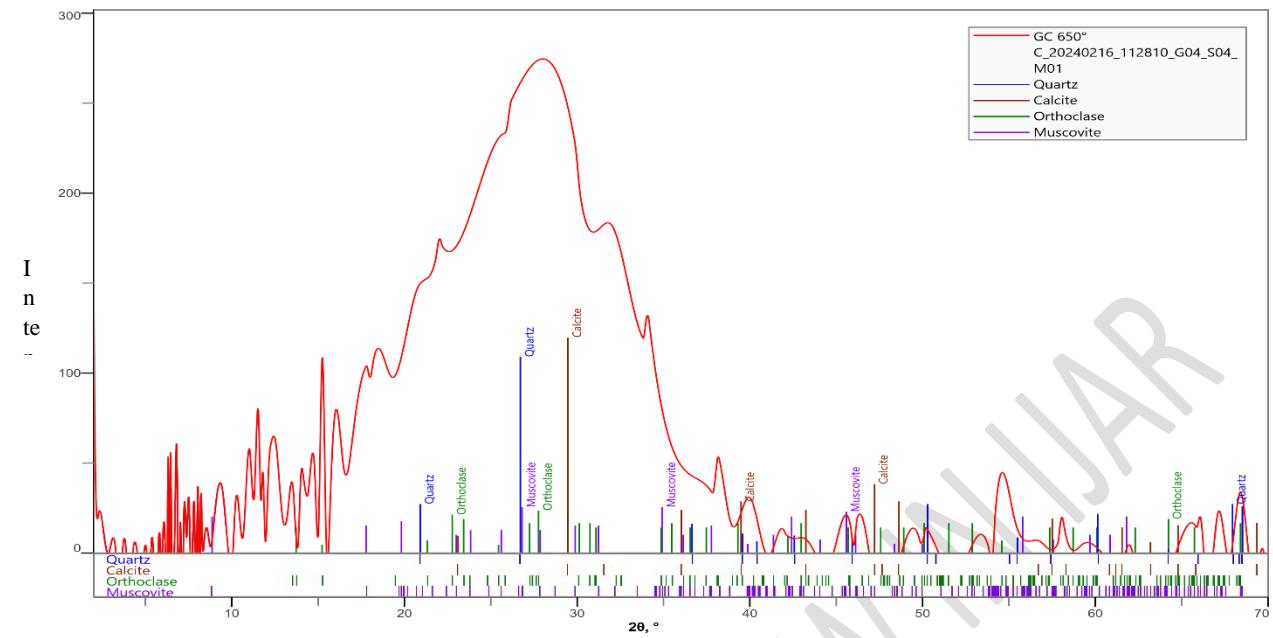
193 **3.3.2 Scanning Electron Microscopy and Energy Dispersive X-ray spectroscopy**
194 (SEM/EDS) the morphology and size of crystals in the glass ceramics were identified by the

195 phenom pro X 800-07334 scanning electron microscope Company Limited, from Netherland. The
196 sinter crystallized glass ceramics demonstrated temperature dependent microstructural evolution.
197 Samples sintered at 650 °C exhibited heterogeneous morphology with open porosity and partial
198 particle bonding. Increasing the sintering temperature from 650 to 750 °C led to enhanced
199 particle coalescence, reduced pore size distribution, and improved densification. This
200 microstructural refinement is directly associated with improved mechanical performance,
201 including increased compressive strength, hardness, and abrasion resistance, confirming the
202 suitability of the developed glass ceramics for tiling applications.



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Plate 1: Sinter Crystallized Glass Ceramics



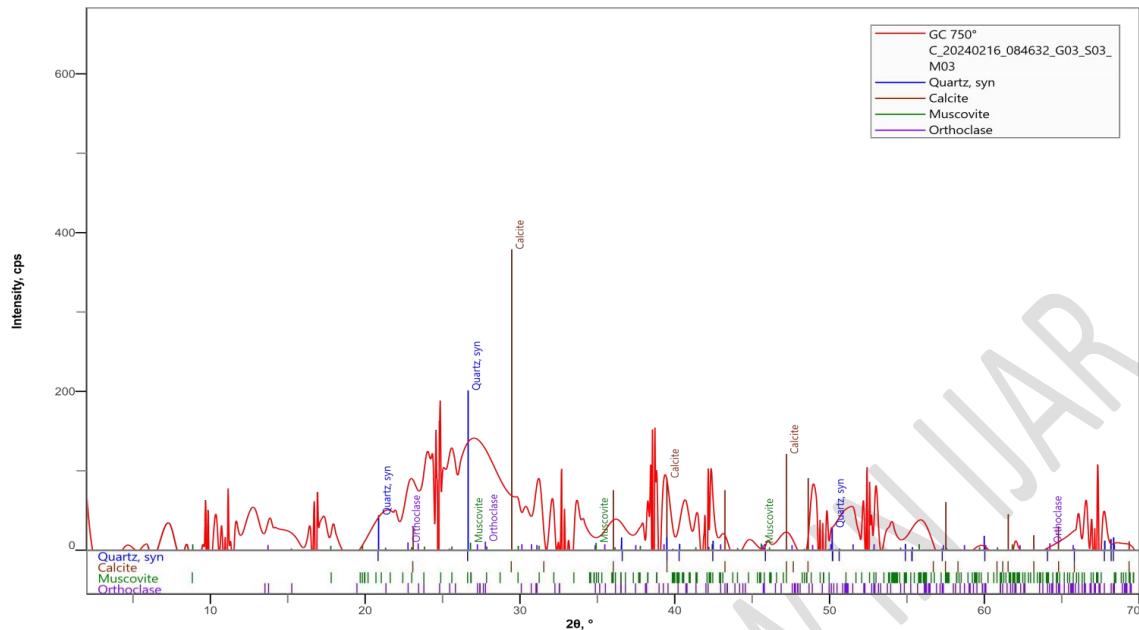
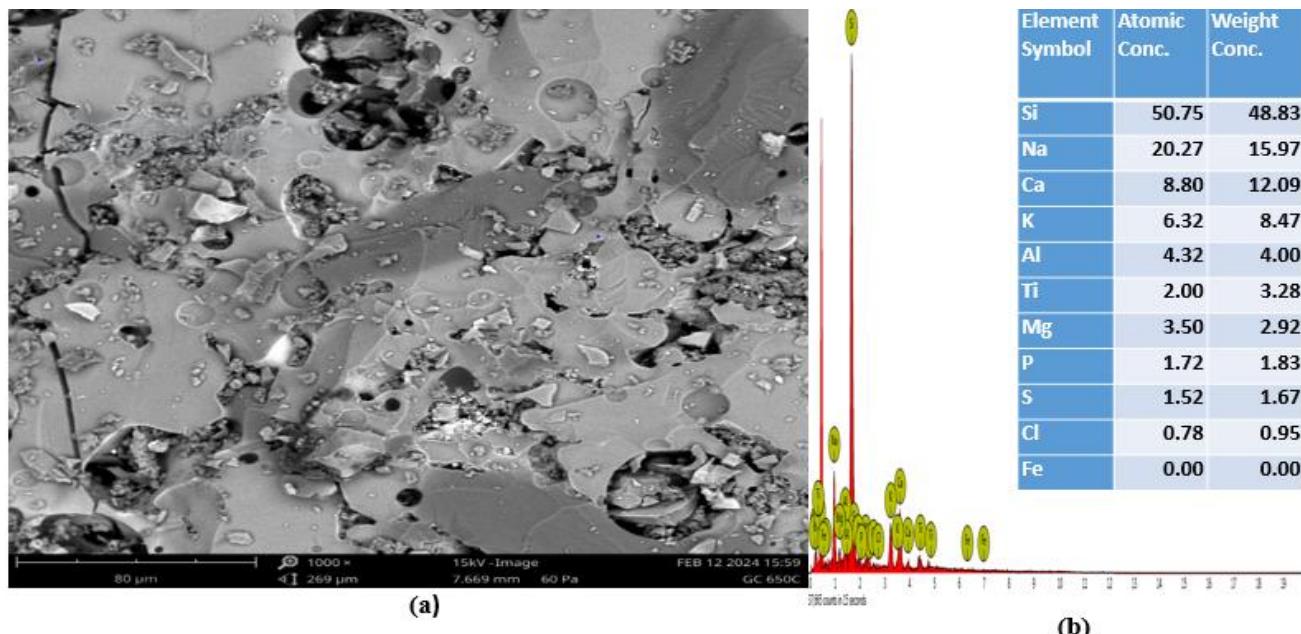


Figure 9: XRD result of Glass Ceramics Sintered at 750 °C

Table 2.Crystallinity index

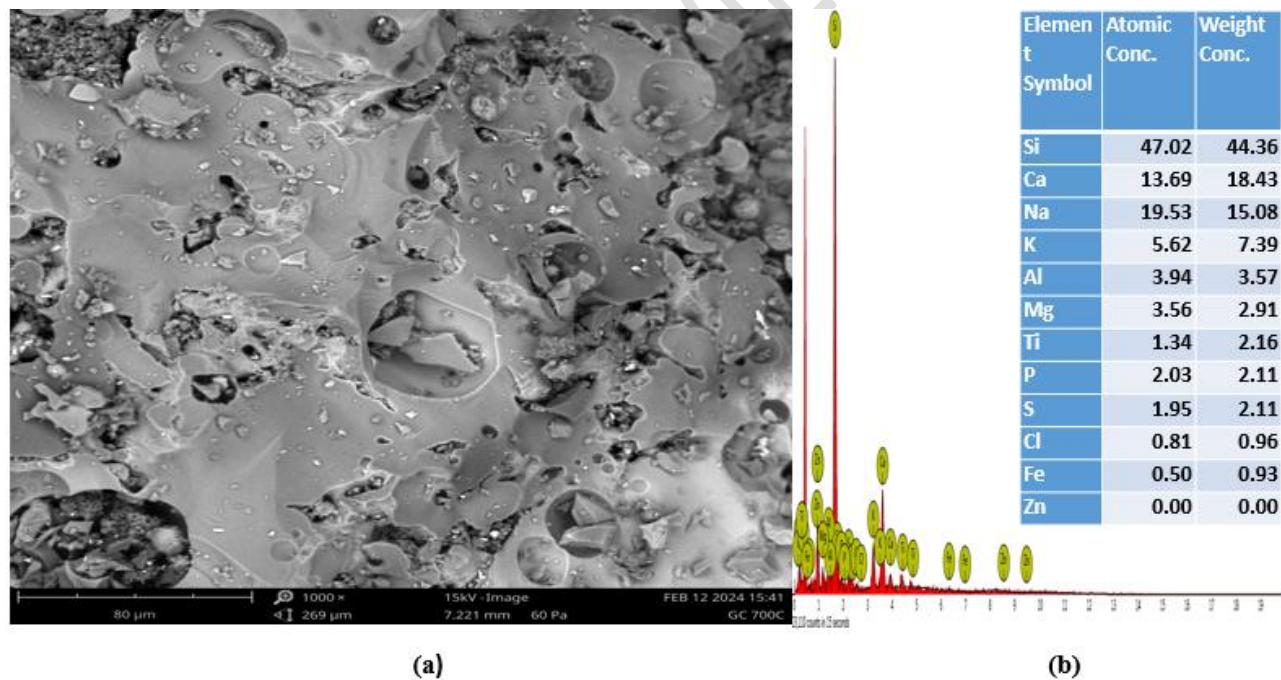
Temperatures		Mineral phases		
°C	Quartz (%)	Calcite (%)	Muscovite (%)	Orthoclase (%)
650 °C	31.7	19	30	19.3
700 °C	35.2	10.7	20.4	33.8
750 °C	44	15	14	27



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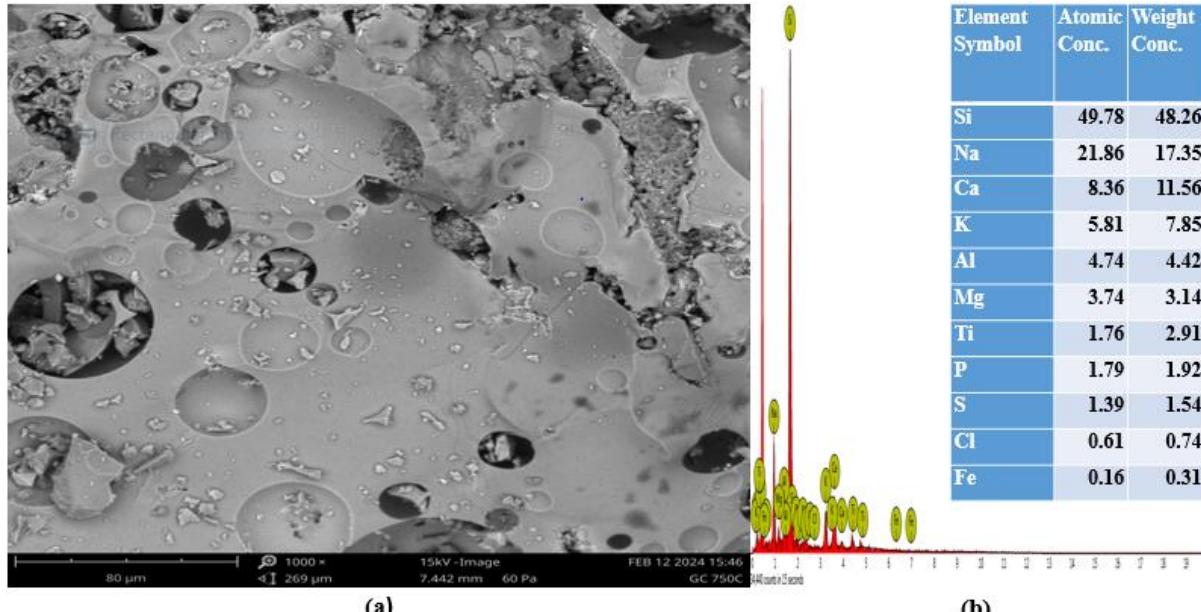
Figure 10: (a) SEM (b) EDS micrographs of Glass Ceramics Sintered at 650°C



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Figure 11: (a) SEM (b) EDS micrographs of Glass Ceramics Sintered at 700°C



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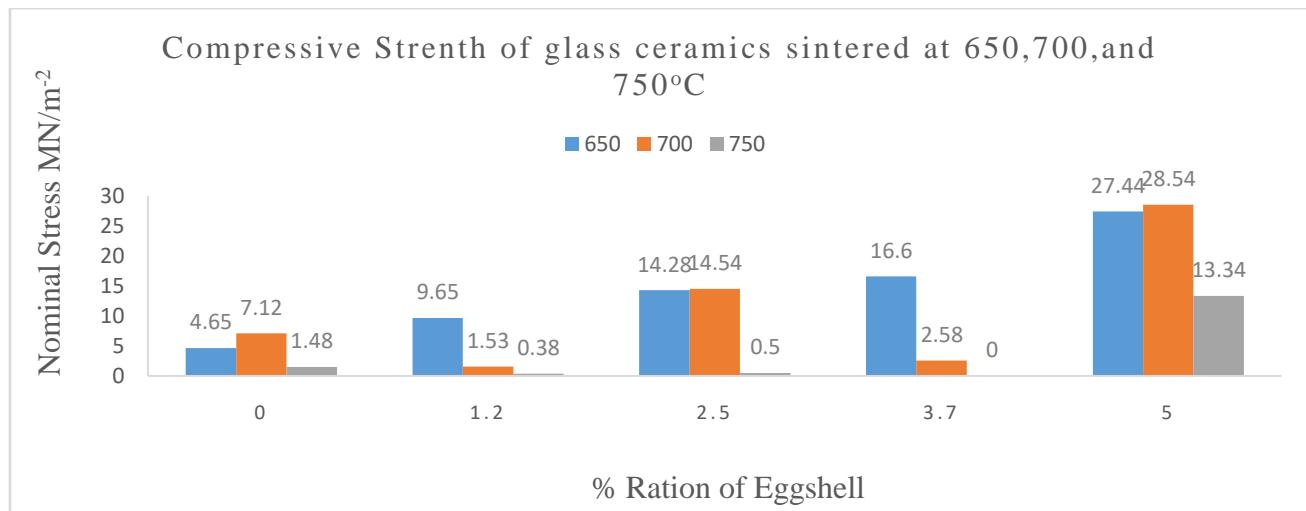
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Figure 12: (a) SEM (b) EDS micrographs of Glass Ceramics Sintered at 750°C

221 3.4. Compressive Strength Test Result

222 The compressive strength of the glass ceramicssintered at 650°C shown in plate Iwas 27.44
 223 MN/m⁻²which increased when the sample sintered at 700°C reaching a value of 28.54 MN/m⁻²,
 224 clearly showing an increase in mechanical strength and also decreases when the temperature
 225 increases to 750 °C reaching a value of 13.34 MN/m⁻²this is attribute to the distortion of shape
 226 and expansion at 750 °C, and result in decreases of mechanical strength. One of the factors that
 227 influence this aspect is the porosity, in which the mechanical properties decrease with the
 228 increase of the porosity according to the generalized mixing rule, aspect correlated and
 229 demonstrated according to the data presented in Figure13, this feature is also associated with a
 230 3.5% increase in the Quartz phase and a 14.5% increase in the orthoclase phase, and 8.3%
 231 decrease in the Calcite phase and a 9.6% decrease in the Muscovite phase. In addition to the
 232 greater cohesion between particles according to the morphology presented in Figure 11, in
 233 general it could be established that the based glass-ceramic materials, depending on the

234 crystallization temperature, present increased mechanical resistance. Therefore, the higher the
235 density, the lower the porosity, and the greater the Compressive strength of the glass ceramics
236 materials.

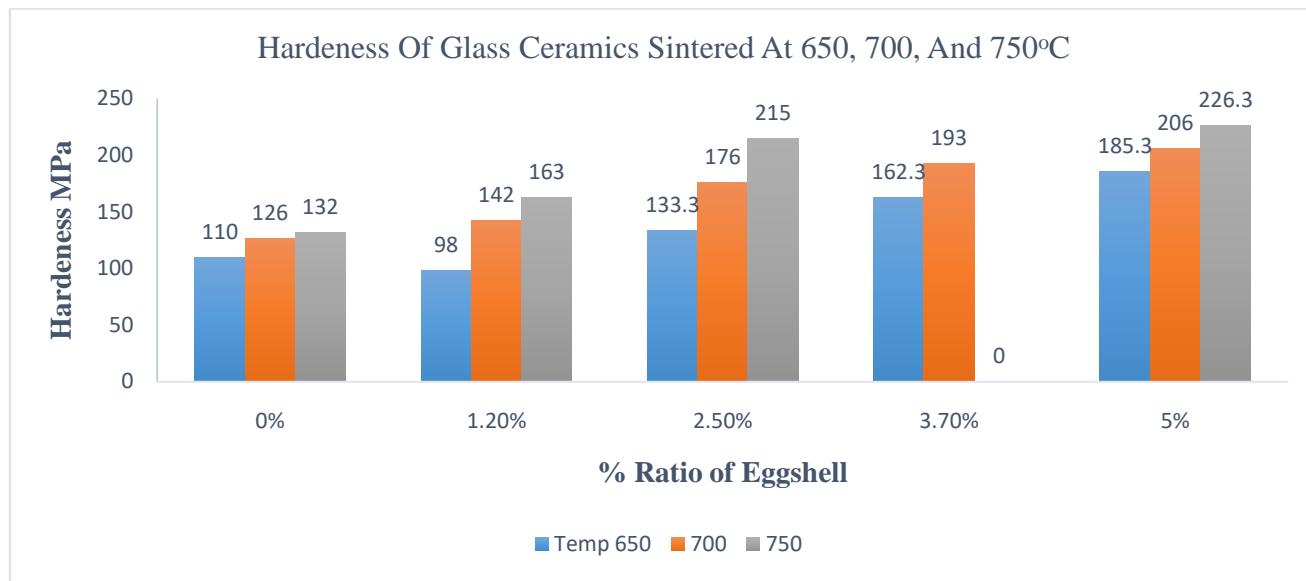


238 Figure 13: Compressive Strength of the sintered glass ceramics

239 **3.5. Vickers Hardness Test Result**

240 The nanoindentation tests results for samples with higher crystal phase contents are shown in
241 Figure 14, Note that the SEM image of the glass ceramics in (Figure 10) showed that these
242 samples are partially crystallized, that is, they contain a residual glassy phase. The calculated
243 hardness values of, glass-ceramic materials are presented in figure 14, the increase in the heat
244 treatment temperature of the glass ceramics provides the highest hardness values. In the case of
245 the Glass Ceramics, the hardness values of the Glass Ceramics heat treated at 750°C with the
246 same 5% Eggshell content are similar and in any case higher than the glass ceramics materials
247 that was heat treated at 650°C, probably due to the viscous flow formation as reported by the
248 SEM image resolution in Figure 12, to cause an increase of the local strength of the Glass
249 Ceramics, The slight differences encountered in temperature changes may outcome from their

250 difference microstructures and the crystalline phases present. Large crystal size and intertwined
251 structures lead to high hardness values.



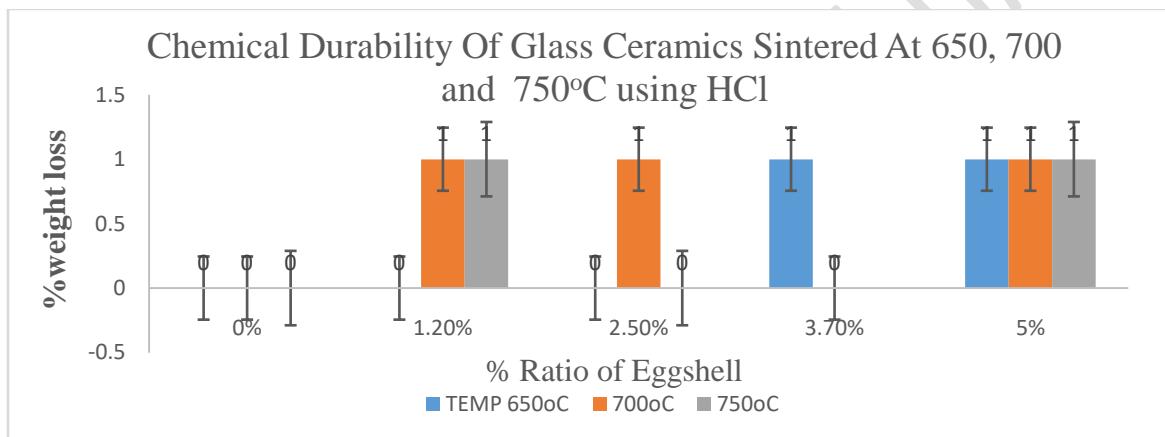
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253 Figure 14: Hardness of the Sintered Glass Ceramics

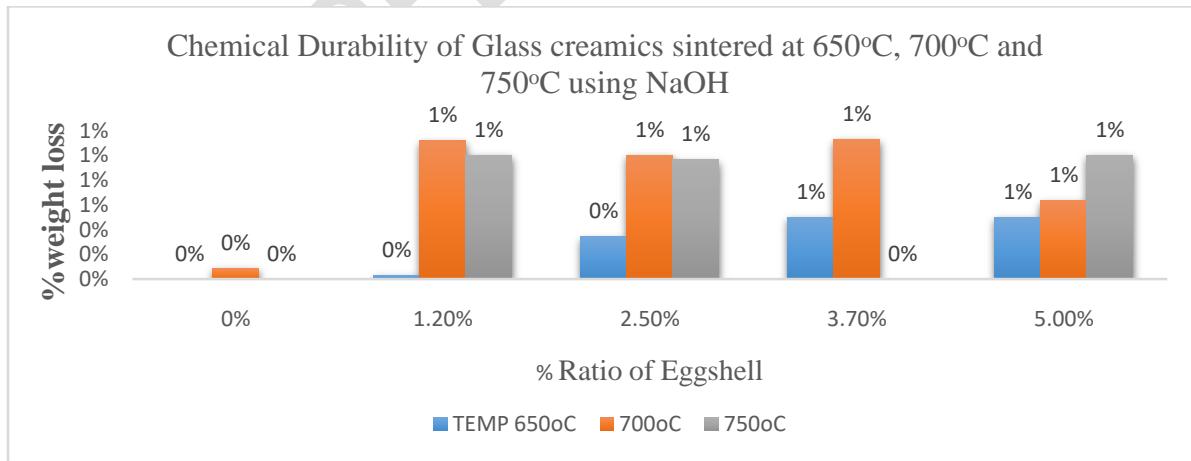
254 **3.6. Chemical durability**

255 In these chemical resistance experiments, 14 compacted sample units with sizes of 25 mm
256 diameter by 11 mm thickness were treated by immersing them in a 1M HCl solution and 1M
257 NaOH solution for 24 hours with a drying process at 130°C for 4h. After washing and drying, the
258 samples were weighed and the percentages of weight loss were calculated by taking the initial
259 mass (m_o) and final mass (m_f) of each of the materials. It was determined using an analytical
260 digital weighing balance with a measurement error of ± 0.0001 g. The mass loss obtained was
261 insignificance after carrying out the previous process to 14 samples, which shows the durability
262 of the sintered glass ceramic samples at 650, 700 and 750°C for 1 hour; this correlates with the
263 volume of the crystalline phases and shows excellent chemical resistance behavior for both
264 samples. The achievement of a high chemical durability in glass ceramics indicates that the

265 chemical composition of the crystalline phase's constituents, composition and amount of residual
 266 glassy matrix favors a good stability. Generally, glass ceramic materials have good chemical
 267 stability and often compare favorably with other ceramic type materials. An increase in the
 268 content of the crystalline phase results in greater chemical resistance in glass ceramic materials.
 269 The obtained result were in agreement with Davalos *et al.*, (2020) who obtained 0.540, as the
 270 highest mass loss for chemical durability index.



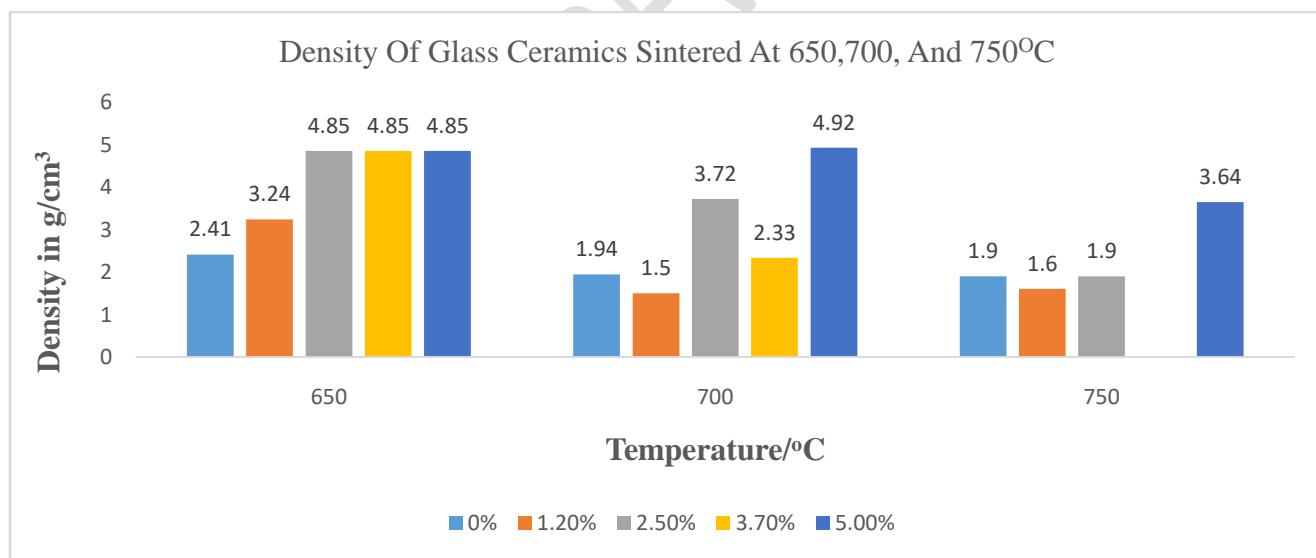
272 Figure 15: Chemical Durability of Sintered Glass Ceramics using HCl



274 Figure 16: Chemical Durability of the sintered Glass ceramics in NaOH solution

275 **3.7. Density of the glass ceramics**

276 The averages density values obtained from five samples of each material. In samples GC650°C
277 and GC700°C, the values of the density are higher in comparison to the sample GC750°C, which
278 can be correlated with a decreased content of crystalline phases in the glass ceramics samples
279 according to the data obtained from the Degree of crystallinity. The decrease in density values
280 from each sample could be correlated with the amount of orthoclase phase present in the material
281 (GC605°C: 19.3%, GC700°C: 33%, and GC750°C: 27%), therefore, it is expected that the
282 density increases or decreases could be as a result of increase or decrease in the crystallinity
283 content. The obtained result is in line with the findings of Valderrama *et al.*, (2021) who
284 conducted a research on Glass-Ceramic Materials Obtained by Sintering of Vitreous Powders
285 from Industrial Waste: Production and Properties.



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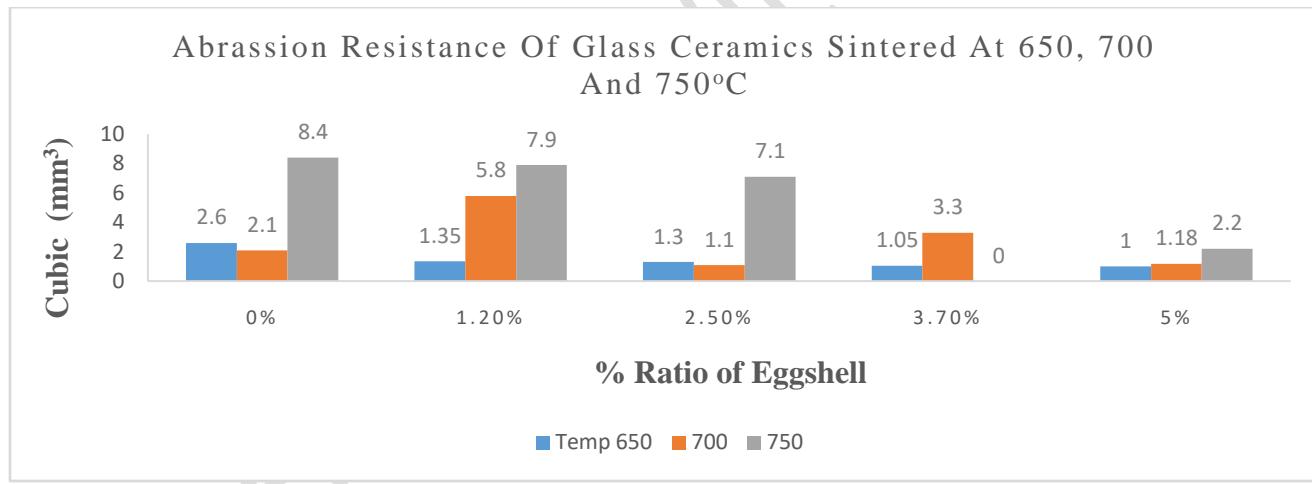
287 Figure 17: Density of the sintered glass ceramics

288 **3.8.Abrasion Resistance**

289 Figure 18 present the abrasion resistance test result. It was observed that sample sintered at 700°C
290 account for higher resistance to scratching and the attributing factor is that during sinter

291 crystallization process, the glassy phase partially crystallizes, forming a network of interlocking
292 crystals within the glass matrices. These crystals can be very hard and have high scratch
293 resistance, contributing significantly to the overall abrasion resistance of the material. The size of
294 these crystals can also play a role. Generally, finer, more uniformly sized crystals can lead to
295 better abrasion resistance compared to larger, more unevenly sized crystals. The integration of
296 high-scratch, hard crystals comprising both glassy and crystalline phases contribute to this
297 resistance. The size of crystals influences their resistance, with finer and uniformly sized crystals
298 demonstrating superior durability. Abrasion resistance correlates directly with density and
299 compressive strength.

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Figure 18: Abrasion Resistance of the Sintered Glass ceramics

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305 ➤ Waste soda lime silicate glass and eggshell derived CaO were successfully utilized to
306 produce glass ceramic materials via the sinter crystallization technique, demonstrating an
307 effective and sustainable waste valorization route.

308 ➤ Chemical and phase analyses confirmed that silica rich waste glass provided the
309 vitrifying network, while eggshell derived CaO acted as an efficient network modifier
310 that promoted controlled crystallization and microstructural refinement.

311 ➤ XRD results revealed temperaturedependent phase evolution, with quartz and calcite as
312 the dominant crystalline phases embedded in a residual glassy matrix.

313 ➤ Increasing sintering temperature enhanced densification and crystallinity up to 700 °C
314 however, excessive heating at 750 °C caused microstructural distortion, increased pore
315 size, and reduced mechanical performance.

316 ➤ SEM observations showed improved particle bonding, reduced pore size, and a more
317 homogeneous microstructure at intermediate sintering temperatures.

318 ➤ The physical, chemical, and mechanical properties of the produced glass ceramics were
319 found to be excellent, therefore, the glass ceramics can be used in tilling application base
320 on the specification provide by the (ASTM C 373 standard that the ceramics tiles should
321 have a minimum water absorption $3\% < E \leq 10\%$, and excellent chemical durability.

322 ➤ The optimum overall performance was achieved at a sintering temperature of 700 °C,
323 which exhibited the highest mechanical strength, density, and abrasion resistance.

324 ➤ The findings establish eggshell derived CaO as a viable alternative calcium source and
325 confirm the suitability of the produced glass ceramics for tile and construction
326 applications, supporting circular economy and sustainable materials development.

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